

(19) World Intellectual Property Organization
International Bureau



(43) International Publication Date
21 August 2003 (21.08.2003)

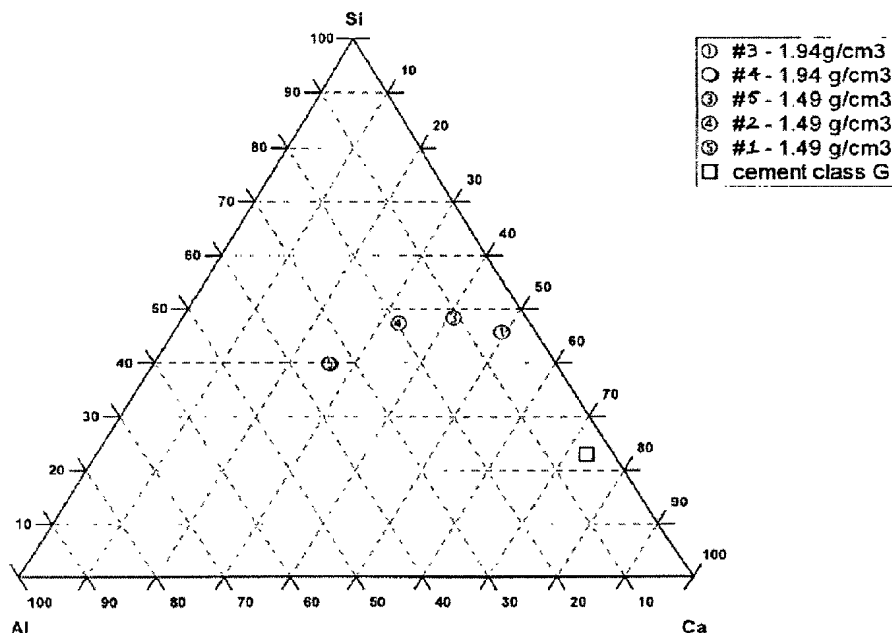
PCT

(10) International Publication Number
WO 03/068708 A1

- (51) International Patent Classification⁷: **C04B 40/00**
- (21) International Application Number: PCT/EP03/01578
- (22) International Filing Date: 14 February 2003 (14.02.2003)
- (25) Filing Language: English
- (26) Publication Language: English
- (30) Priority Data:
0203729.9 18 February 2002 (18.02.2002) GB
0213690.1 14 June 2002 (14.06.2002) GB
0301060.0 17 January 2003 (17.01.2003) GB
- (71) Applicant (for FR only): **SERVICES PETROLIERS SCHLUMBERGER** [FR/FR]; 42, rue Saint Dominique, F-75007 Paris (FR).
- (71) Applicant (for AT, AZ, DE, DK, IT, KP, KR, KZ, NO, NZ, PL, RO, RU, TM, UA only): **SCHLUMBERGER TECHNOLOGY B.V.** [NL/NL]; Parkstraat 83-89, NL-2514 JG The Hague (NL).
- (71) Applicant (for GB, JP, NL only): **SCHLUMBERGER HOLDINGS LIMITED** [—/—]; P.O. Box 71, Craigmuir Chambers, Road Town, Tortola (VG).
- (71) Applicant (for all designated States except AT, AZ, CA, DE, DK, FR, GB, IT, JP, KP, KR, KZ, NL, NO, NZ, PL, RO, RU, TM, UA, US): **SOFITECH N.V.** [BE/BE]; 140, rue de Stalle, B-1180 Brussels (BE).
- (71) Applicant (for CA only): **SCHLUMBERGER CANADA LIMITED** [CA/CA]; 24th Floor, Monenco Place, 801 6th Avenue SW, Calgary, Alberta T2P 3W2 (CA).
- (72) Inventors; and
- (75) Inventors/Applicants (for US only): **BARLET-GOUEDARD, Véronique** [FR/FR]; Résidence La Baleine, 3, rue du Mérou, F-92290 Chatenay Malabry (FR). **DANICAN, Samuel** [FR/US]; 2500 Old Farm Road, Houston, TX 77063 (US). **NELSON, Erik** [US/US]; Schlumberger, P.O. Box 4610, Sugar Land, TX 77210-4610 (US). **GOFFE, Bruno** [FR/FR]; 11 bis Rue Larrey, F-75005 Paris (FR). **CAMBUS, Chrystel**

[Continued on next page]

(54) Title: CEMENT COMPOSITIONS FOR HIGH TEMPERATURE APPLICATIONS



(57) Abstract: In a process to design high temperature cement slurries, the temperature stable phases (anorthite, wairakite) are selected; aluminum modifiers and silica oxides are added to the dry cement so that the elemental composition of the blend corresponds to the phases selected; the control of the particle size distribution and the relative amount of those minerals allow their addition at high concentration while leaving the slurry easily mixable and pumpable; and the kinetics of the targeted phases formation is then controlled by adjusting the crystallinity and the particles sizes of the different solids.

WO 03/068708 A1



[FR/FR]; Le Four à Chaux, Route de Houdan, 78550 Richebourg (FR).

(74) **Agent: HYDEN, Martin**; Etudes & Productions Schlumberger, 1, rue Henri Becquerel - BP 202, F-92142 Clamart Cedex (FR).

(81) **Designated States (national)**: AE, AG, AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, BZ, CA, CH, CN, CO, CR, CU, CZ, DE, DK, DM, DZ, EC, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MA, MD, MG, MK, MN, MW, MX, MZ, NO, NZ, OM, PH, PL, PT, RO, RU, SC, SD, SE, SG, SK, SL, TJ, TM, TN, TR, TT, TZ, UA, UG, US, UZ, VC, VN, YU, ZA, ZM, ZW.

(84) **Designated States (regional)**: ARIPO patent (GH, GM, KE, LS, MW, MZ, SD, SL, SZ, TZ, UG, ZM, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, BG, CH, CY, CZ, DE, DK, EE, ES, FI, FR, GB, GR, HU, IE, IT, LU, MC, NL, PT, SE, SI, SK, TR), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GQ, GW, ML, MR, NE, SN, TD, TG).

Published:

— with international search report

For two-letter codes and other abbreviations, refer to the "Guidance Notes on Codes and Abbreviations" appearing at the beginning of each regular issue of the PCT Gazette.

CEMENT COMPOSITIONS FOR HIGH TEMPERATURE APPLICATIONS

The present invention relates to cement compositions for high-temperature applications. The invention provides not only compositions, but methods for the design of such compositions and methods for their use, particularly in relation to the cementing of wells or pipes.

In the construction of wells, cement is used to secure and support casing inside the well and prevent fluid communication between the various underground fluid-containing layers or the production of unwanted fluids into the well. Such techniques are used when completing underground oil, water, gas or geothermal wells or the like. In geothermal applications, cement is also often used to line pipes carrying the steam or hot brine from the well to a location where it can be used for heating or power generation. Such applications constitute particularly aggressive environments for cement, both by virtue of the high temperature and the frequent presence of aggressive chemical environments such as brines and carbon dioxide. In order that the cement can fulfil the functions set forth above, it is necessary that the cement is as durable and resistant to chemical attack as possible while still being able to perform the desired functions. The behaviour of cements under such conditions is discussed in Berra, M. et al. (1988), Behaviour of cementing hydraulic binder under severe geothermal conditions. Geothermics, Vol. 17, No. 5/6, pp 785 – 813.

In most cases, the hydraulic binder used in such cementing applications is Portland cement, often with the addition of various solid and liquid additives. Portland cement is composed of calcium silicates and calcium aluminosilicates. In the presence of water, those mineral phases hydrate and form hydrated calcium silicate (CSH) and also portlandite ($\text{Ca}(\text{OH})_2$) and aluminium sulphate. At temperatures below about 100°C , CSH is a gel, and is largely responsible for the cohesion and mechanical strength of the set cement in the short to medium term.

When the temperature exceeds about 100°C , in the first days or months the majority of cements lose compressive strength and become permeable. Then they regain their strength and their permeability reduces. That phenomenon, known as retrogression, is

due to metamorphosis of CSH gel to a highly crystalline phase known as alpha dicalcium silicate, α -C₂SH, which is much more dense than CSH with a contraction in volume and an increase in the porosity and permeability of the cement. This results in disintegration of the cement, which increases in rate as the porosity and permeability increases, rendering it more sensitive to chemical attack such as channelling of brine or circulation of carbon dioxide.

To avoid the phenomenon of retrogression, it has been proposed to add silica in the form of sand or silica flour to compositions which have to withstand temperatures of more than 100°C. Adding 35% to 40% of silica (with respect to the weight of the Portland cement) brings the CaO/SiO₂ ratio (usually denoted C/S) to about 1. An example of such an approach can be found in WO 97/49644.

With this modified composition, CSH gel reacts with the added silica to form the mineral tobermorite (C₅S₆H₅) at about 110°C. When the temperature is raised to 150°C, the tobermorite transforms into xonotlite (C₆S₆H). These minerals are intrinsically strong and can improve the stability and mechanical integrity of the cement at high temperatures in the absence of brine; however, owing to slight differences in their respective densities, the recrystallizations can destabilise the cement and cause the permeability of the set cement to increase. At 250°C truscottite (C₇S₁₂H₃) begins to appear. Owing to the manner in which truscottite crystallizes, a reduction in permeability is usually observed. (Nelson, Well Cementing - "Thermal Cements" 1990 9.1 – 9.19; J P Gallus et al, "Performance of Oil-Well Cementing Compositions in Geothermal Wells" SPE 7591 1978; L H Eilers et al, "High Temperature Cement Compositions – Pectolite, Scawtite, Truscottite or Xontolite: Which Do You Want?" JPT 1983 1373-1377). As the temperature approaches 400°C (750°F), both xonotlite and truscottite are near their maximum stable temperatures, and dehydration of residual CH to C occurs (Nelson, Well Cementing – "Thermal Cements" 1990 9.2)

The degradation mechanisms described above are certainly not exhaustive, the roles of other oxides present in the Portland cement, in particular alumina and ferrites, have not been presented, but they suffice to demonstrate the complexity of the phenomena

occurring, without even considering the reaction products between the cement matrix and brines. In nature, in a geological medium, cement is thus meta-stable. In contact with brine, its mineralogical repertoire is modified by ion exchanges, which modify its physical qualities.

5

The problems of cement disintegration when the temperature is raised are more severe when developing very deep wells, injecting steam to recover highly viscous oils (steamflood), in geothermal, or in in-situ combustion recovery applications (firefloods). Geothermal wells are wells drilled into formations that contain

10

superheated brines. At surface the brine is converted to steam which is then used to produce electricity. Steamflooding is a process where steam is injected into formations that contain viscous oil. The rise in temperature causes the viscosity of the oil to reduce, allowing it to flow to a production well and to be pumped to surface.

15

The temperature and pressure conditions encountered in such wells vary between 200–300°C and 10 MPa – 95 MPa, respectively. Fireflood consists of initiating combustion underground in an injection well, and then propagating the combustion front through the underground formation by the injection of air through the reservoir to the production well. In such wells, the cement is exposed to maximum temperatures between 371°C and 926°C (700°F – 1700°F) near the burning zone. Such temperatures exceed the stable range of Portland cement.

20

One approach explored in work by Roy et al. (“Potential New High Temperature Cements for Geothermal Wells” AIME International Symposium on Oilfield and Geothermal Chemistry, Houston, TX 1979, pp 153-161) showed that the mineral anorthite was promising as a binder in geothermal well cements. However, the practical problem of formulating a pumpable cement slurry with a predictable working time remained to be solved. GB 1,525,236 and JP 3055899 both disclose the formation of anorthite in the manufacture of cementitious materials but give no teaching on the relevance of such a mineral phase to the high-temperature cementing operations of the present invention.

25

30

EP 0 922 013 A proposes compositions which may be stable at the temperatures of interest which can promote the formation of calcium hydrogarnets from the series

[SiO₄]Ca₃ M³⁺, where M = Al (grossularite) or M = Fe (andradite) by adding alumina or ferrite. The quantity added is typically in the range 20% to 50% (by mass with respect to the mass of Portland cement), i.e., of the same order of magnitude as the amount of silica added. A more particularly preferred composition is constituted by class G Portland cement, 30% to 40% of silica (preferably of the order of 35%) and 20% to 40% of alumina (preferably of the order of 30%). The mineral phases favoured by these compositions are principally of varying composition, in particular hydrogarnets from the hydrogrossularite series, with a structure derived from calcium garnets Ca₃Al₂Si₃O₁₂ (C₃AS₃) with the silica atoms omitted and replacement of the oxygen atoms to which they were bonded by hydroxyl groups. When all of the silicon atoms are replaced, C₃AH₆ is obtained. Under the selected experimental conditions (288°C, 20 MPa), those garnets have been shown to be poorly constituted in part, with low reproducibility of crystallisation leading to larger or smaller crystals associated with high heterogeneity in compression test results.

WO01/70646 A proposes formulations suitable for cementing oil, gas, geothermal wells or the like, based on a Portland and alumina type cement tending to form mineral phases such that the properties of the cement matrix, in particular regarding compressive strength, do not degrade in practice under the effect of a rise in temperature and pressure (250°C-300°C; 20 MPa) and/or chemical attack (brines). The proposed compositions provide a cement matrix included in the Si-Ca-Al triangle in one of the margarite-häüyne-[epidote/pumpellyite], häüyne-prehnite-[epidote/pumpellyite] and häüyne-prehnite-pectolite composition triangles. In these triangles of the Si-Al-Ca-Na system, many mineral phases exist that are stable under the temperatures and pressures under consideration. The preferred compositions of WO01/7064 are located in the most calcium-rich zones, in particular close to the margarite-[epidote/pumpellyite]-pectolite junctions and margarite-prehnite-pectolite junctions, corresponding to compositions within the Si-Ca-Al triangle approximately in the zone delimited by silicon between 35% and 50%, aluminium between 20% and 38% and calcium between 25% and 35%. The adjustment of the aluminium content of the compositions is achieved by the use of micro-alumina. While the correct mineral content is achieved by the use of this material, problems remain due to its relatively low reactivity and high cost.

Other approaches to the problems of high temperature cements can be found in US 4,642,137; US 4,877,452; US 4,935,060; US 5,158,613; US 5,900,053; US 6,143,069; US 6,332, 921; US 6,367,556; and US 6,488,763.

- 5 EP 0 621 247 A describes a cement composition formed from particles having controlled particle size ranges or granulometries. By optimising the packing volume fraction of the particulate materials, cement slurries can be designed that have improved properties while remaining pumpable in normal field conditions.
- 10 It is an object of the present invention to provide a method to allow the design of cementing compositions suitable for the high-temperature applications discussed above with the aim of providing pumpable slurries that have sufficient strength and durability for the desired use. It is also an object to provide a series of such compositions and methods for their use.
- 15 A first aspect of the present invention provides a method of designing a cement slurry, comprising determining a temperature to which the cement slurry will be exposed in situ; determining a stable, thermodynamic equilibrium composition of a CaO-Al₂O₃-SiO₂-H₂O (CASH) mineral system at the determined temperature; determining
- 20 proportions of cement and mineral oxides required to provide a mixture having the determined composition of the CASH mineral system; and defining a series of particulate materials of predetermined particle sizes and densities, and comprising the cement and mineral oxides in the determined proportions such that, when mixed with water, a pumpable slurry is formed.
- 25 This aspect of the present invention resides in the recognition that rocks of basic compositions (poor in silicon, rich in calcium) maintain low porosity and a good mechanical resistance during weathering and can be considered as good natural analogues to conventional cements. Anorthite, grossular, prehnite and zeolite occur in
- 30 basic rocks of magmatic origin, transformed in metamorphic or hydrothermal conditions. The method according to the invention allows the development of formulations suitable for cementing oil, gas, geothermal wells or the like, based on a Portland and alumina type cement tending to form mineral phases corresponding to

these basic rock phases such that the properties of the cement matrix, in particular regarding compressive strength, do not degrade in practice under the effect of a rise in temperature and pressure (250°C-900°C; 20 MPa) and/or chemical attack (brines, CO₂). By determining the chemical composition of a stable mineral system at the temperature of interest, and ensuring that the cementing composition has this composition, it is possible to provide a set cement that is stable, even at very high temperatures such as are encountered in fireflood wells.

The proportions of cement and mineral oxides required to form a stable cement can be determined from general thermodynamic rules, phase relations between minerals and fluids and minerals, activity models and homogenous and heterogeneous phase equilibria known and developed for natural silicon- and calcium-rich rock systems of mafic origin to predict the mineralogy and chemical behaviour of Portland cements or derived compounds set in environments comparable to those encountered in geological settings such as hydrothermal fields of metamorphism (temperature greater than 150-200°C, pressure greater than liquid-vapour transition curve).

The method provides cement slurries that promote early anorthite formation during the hydration of operating slurry by adding specific modifiers. Anorthite, characterized by solid solutions, exhibits the best adaptation to severe chemical environment.

In order to promote the formation of anorthite, specific modifiers have to be added to the cement powder. At the same time, the final slurry density often has to be as low as possible (e.g. 1.4 to 1.5 sg) (due to the low fracture gradients generally experienced in geothermal wells) while maximizing the solid fraction of the slurry to decrease the permeability and increase the compressive strength. Higher densities (e.g. 1.9 sg) may also be provided for other applications such as fireflood.

To achieve this, the physics of optimizing the packing between the different components of the material has to be mastered. The particle size distribution of each component has to be optimum. For example, the solid components can be provided in three or four distinct particle size fractions to optimize the amount of solids in the

mixture. This technique allows the use of large amounts of solids while making the slurry still easily pumpable.

Class G cement can conveniently be used as the hydraulic binder. Cenospheres or
5 other hollow particulate material of very low density (0.7 to 0.8 sg) can be added to decrease the slurry density. A particularly preferred material in this class is S060 from the 3M Company which comprise high strength, low density microspheres having a particle size range of 50 – 300 microns and a density of $0.60 \pm 0.06 \text{ g/cm}^3$. Such materials are particularly useful in high pressure applications. Some of these materials
10 can also add silica or aluminum to the systems in certain circumstances, depending on their composition and form. Silica, silica flour and aluminum modifiers (alumina, micro-alumina, clay as kaolinite and calcined products of kaolinite) can comprise other components of the blends.

15 Apart from the cement for which the particle size is typically defined by the manufacturer, particle size of cenospheres, silica and aluminum modifiers are carefully selected in order to achieve maximum packing of the blend. The size and the crystallinity of the different blend components allow control of the kinetics of anorthite formation.

20 A cementing composition according to the first aspect of the invention comprises Portland cement, a silica modifier comprising a material that contributes silica to the composition during setting reactions, and an aluminium modifier comprising a material that contributes aluminium to the composition during setting reactions,
25 characterized in that the cement, silica modifier and aluminium modifier together define a stable, thermodynamic equilibrium composition of a $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2\text{-H}_2\text{O}$ (CASH) mineral system at the temperature at which the cementing composition is to be placed, the cement, silica modifier, aluminium modifier and any other particulate additives defining a series of particulate materials in predetermined, substantially non-
30 overlapping particle size ranges, and water in sufficient quantity to form a pumpable slurry.

In accordance with the present invention, there is provided a cement composition comprising a mixture of: (i) an hydraulic cement; and (ii) one or more additives that contribute Si, Ca and Al to the mixture; characterized in that it has a mineralogical composition within the Margarite-Quartz-Pectolite region of the Si-Ca-Al phase diagram, and in that the constituents are selected to optimise the formation of Anorthite.

Such compositions are useful for cementing wells such as oil, gas, geothermal, and water wells, in particular such wells that are subjected to high temperatures (in excess of 250°C. Another application for such cements is to serve as an inner lining for pipelines, particularly those that carry brines produced from geothermal wells. The cement lining prevents corrosion of the pipe as the brine is transferred from the wellhead to the power generating station.

Specific modifiers are added to cement powder to promote the formation of Anorthite. At the same time, it is preferred that the density range be as large as possible (for example, 1.3 sg to 1.94 sg) while maximizing the solid fraction of the slurry to decrease the permeability and increase the compressive strength. It is also preferred that the slurries are stable with respect to sedimentation.

To achieve these goals, the packing between the different components of the blend is optimised by selecting the particle sizes of the cement and additive materials used to make the slurry. By controlling the particle sizes of the solid constituents of the slurries, the packing volume fraction can be optimised. This technique allows the use of large amounts of solids while making the slurry still easily pumpable and stable with respect to sedimentation.

In accordance with a second aspect of the present invention, there is provided a cement composition comprising a mixture of: (i) an hydraulic cement; and (ii) one or more additives that contribute Si, Ca and Al to the mixture; characterized in that at least one additive comprises, kaolin, calcinations products of kaolin, metakaolin or calcinations products of metakaolin, in that it has a mineralogical composition within the Margarite-Quartz-Pectolite region of the Si-Ca-Al phase diagram, and in that the constituents are selected to optimise the formation of Anorthite.

Compositions falling within the Margarite-Quartz-Pectolite region of the Si-Ca-Al phase diagram have been found to be particularly stable at high temperatures. Furthermore, selecting the constituents to optimise the formation of anorthite avoids the formation of phases that, while stable at high temperatures, have a negative impact on the structural stability of the set cement, e.g. xonotlite in presence of brine or CO₂.

It is particularly preferred that the additives are selected to have a high reactivity such that anorthite forms quickly so as to avoid extensive formation of xonotlite. The chemical composition can be selected considering the selected Si-Ca-Al ratio, to optimise total reactivity and fast reactivity of the blend at a given temperature and pressure and the optimised particle size distribution in accordance with the optimised packing volume fraction.

Preferably, the compositions of the invention are located in the Si-Ca-Al triangle, in the zone delimited by silicon between 44% and 57%, aluminium between 18% and 32% and calcium between 18% and 32%.

Compositions in accordance with the present invention can be realised using different materials containing silicon, and/or aluminium which are added to an oilwell cement with or without pozzolanic material. The reactivity and particle size of each material are selected in order to quickly form anorthite and to reach a good mixability respectively.

Microsilica and silica/alumina microspheres, metakaolin (calcined kaolinite), kaolinite (clay), fly ash and silica with different particle sizes can be used as silicon and aluminium sources.

Calcination products of kaolin and/or metakaolin are particularly preferred materials. Calcination products of metakaolin comprise metakaolin that has been heated such that all water has been lost by dehydroxilization and at least part of the metakaolin phase has decomposed to spinel-type phase. Such compositions can be achieved by heating (calcining) metakaolin above 750°C. The degree of calcining (temperature and duration of heating) can be selected according to specific requirements. Full

calcinations or sintering may not be appropriate in some circumstance. Such products have the advantage that they have good reactivity so as to form anorthite quickly and avoid the formation of xonotlite.

- 5 The present invention will now be described by way of examples, with reference to the accompanying drawings, in which:
- Fig. 1 shows a plot of blend compositions in the $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2$ ternary diagram;
- Fig. 2 shows mineral phases known to be stable in natural calcium-rich systems plotted of in the $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2$ ternary diagram (H_2O in excess);
- 10 Figs. 3a and b show SEM microphotographs of set cement matrix obtained in a “xonotlite dominant” sample;
- Figs. 4a and b show SEM microphotographs of set cement matrix obtained in an “anorthite dominant” sample;
- Figs. 5a and b show SEM microphotographs of a set cement matrix of a further
- 15 sample;
- Figures 6 – 14 show photographs of cement samples after ageing at high temperatures for varying durations; and
- Figure 15 shows the phase diagram of a ternary silicon/aluminium/calcium system showing the blend compositions tested (1 – 9) and various mineral phases.
- 20 Different cement blends are prepared according to the method of the invention. Table 1 summarizes the physical properties of the blends and slurries. All slurry properties have been measured as per API specifications (API Recommended Practice for testing Oil-Well Cements and cement Additives (1990) API Spec.10 Fifth Edition
- 25 (07.01.1990)). The compositions of blends in the quaternary system ($\text{CaO-SiO}_2\text{-Al}_2\text{O}_3\text{-H}_2\text{O}$) are shown in Figure 1. For comparison purposes, composition #1 is made according to prior art techniques.

Table 1: Properties of the slurries

No. slurry	#1	#2	#3	#4	#5
Density (g.cm^{-3})	1.4	1.49	1.49	1.94	1.94
W/C ratio	1.3	0.58	0.81	1.07	0.69
Solids Volume Fraction	0.40	0.59	0.59	0.43	0.42

Free Water	0	0	0	0	0
Blend	Cement Silica flour Cenospheres -	Cement Silica flour cenospheres Aluminum modifier	Cement Silica flour cenospheres Aluminum modifier	Cement Silica flour Silica Aluminum modifier	Cement Silica Flour - -
Plastic viscosity (cP)/Yield value after mixing (lbs/100 sqft)	33/6	257/29	109/9	53/21	149/11
Plastic viscosity (cP)/Yield value after 20 min conditioning (lbs/100 sqft)	16/13	109/3	54/6	30/15	85/7
Reading on FANN mod.35SA viscosimeter (R1-B1-F1) Gel at 10 min	11	21	15	35	4
Thickening Time (hours)	6:00	4:54	4:16	5:09	4:52
50psi	3:43	4:40	4:34	6:02	-
500psi	11:55	5:53	6:15	6:55	4:29
Transit Time	0:06	0:02	0:05	0:15	0:05
Compressive Strength after 24 hours at 400°F	680	2100	1480	2420	6000

The slurries are mixed following the API procedure. Mixing is done in a Waring Blender. All solid components are dry blended before being added to the water containing the liquid additives. Samples are prepared for curing by pouring the mixed slurry into a 5cm x 5cm x 5cm steel molds. Five cubes are prepared for each formulation.

The molds containing the slurries are cured for three days at 300°C over 21 MPa in a standard oil well cement-curing chamber in water. The temperature is gradually increased to 300°C over a 6-8 hour heat up period, to minimize possible cracking of the samples due to thermal expansion. The temperature is maintained at 300°C and 21 MPa for 28 days, after which the system is allowed to cool down to room temperature.

The cured set cement samples obtained after 28 days exposure at 300°C, are drilled in 2.5 x 5 cm cores. These set samples are exposed to the production brine at 300°C over 21 MPa in the high pressure – high temperature geothermal reactor for 28 days. The brine is regularly changed (every 3 days) to keep the solution saturated and to simulate as close as possible actual well conditions, where the brine is an “open system”. The heating rate to reach 300°C is 6°C/min.

Samples are then removed, kept under water at room temperature before being tested for the following physical property measurements:

- 5 ☐ Uniaxial compressive strength performed on 2.5 x 5 cm cores in using a compressive machine
- ☐ Water Permeability

Mineralogical examination is carried out by x-ray powder diffraction using CuK α radiation. Scanning Electron Microscope images are used to describe morphologies of reaction products and to complete the XRD analysis.

Two types of brines are used to simulate the chemical attack. Table 2 shows the chemical composition of a production brine circulated in a Japanese geothermal well at 1300 meters with a production temperature around 300°C. The pH of this brine is 2.6.

Table 2: Chemical composition of the production brine (Brine1)

Ions	mg/l
Na	3120
K	810
Ca	1460
Mg	270
Cl	9570
SO ₄	201
SiO ₂	144
Fe	648

The table 3 shows the composition of a California brine.

Table 3: Chemical composition of the field brine (Brine 2)

Ions	g/l
NaCl	200
KCl	23
CaCl ₂	63
MgCl ₂ .6H ₂ O	1

The results of the mineralogical composition of the different samples before and after brine attack are presented in Table 4 below:

5 Table 4: Mineralogical composition of the different samples before and after brine attack

Formulation Number	#1		#2		#3	#4		#5	
Density g/cm ³	1.49		1.49		1.49	1.94		1.94	
Brine 1 or 2 attack at 300°C @28days	Before brine attack	After Brine 1 attack	Before brine attack	After Brine 1 attack	Before brine attack	Before brine attack	After Brine 2 attack	Before brine attack	After Brine 2 attack
Xonotlite C ₆ S ₆ H	+++ +	+++ +	++	++	++	++	++	+++ +	+++ +
Anorthite CaAl ₂ Si ₂ O ₈	++	++	+++ +	-	++++	+++	+++ +	-	-
Anorthite,sodium,intermediate (Ca,Na)(Si,Al) ₄ O ₈	-	-	-	+++ +	-	-	++	-	-
Albite NaAlSi ₃ O ₈	-	-	-	+++	-	-	-	-	-
Wairakite CaAl ₂ Si ₉ O ₁₂ .2H ₂ O	-	-	++	++	++	-	-	-	-
Prehnite (zeolithe) [Si ₃ AlO ₁₀]Ca ₂ Al(OH) ₂	-	-	-	-	-	++	+	-	-
Epidote 1-2 [Si ₂ O ₇][SiO ₄](OH)Al ₂ O ₂ CaAl ₂	-	-	-	-	-	++	++	-	-
Garnet [SiO ₄] ₃ Al ₂ Ca ₃	-	-	-	-	-	-	++	-	-
Quartz SiO ₂	+	-	-	-	+	+++	-	-	-
Corundum μAl ₂ O ₃	+	-	+	-	++	++	-	-	-

++++ Prevailing phase; +++ phase in appreciable amount; ++ phase in fairly good amount; + phase in small amount.

Figure 2 presents in the same quaternary system CASH as shown in Figure 1, the
5 blend compositions from Table 1 and the different mineral phases of basic rock after hydrothermal alteration known as stable at 300°C.

Two areas can be defined:

- Area 1, outside the two joins Epidote-pectolite and Prehnite-pectolite
- 10 • Area 2 near or inside these two joins where the formation of xonotlite can be avoided or limited.

In area 1, two compositions have been tested, #1 and #5 with 1.49 and 1.94 g/cm³ densities respectively. The phase responsible for the cementing properties within this
15 region is mainly xonotlite.

With the low-density cement composition (#1), anorthite in a fairly good amount is easily detected. Its composition contains more alumina than the one at higher density. After brine 1 attack, the set cement matrix shows deleterious signs and the water
20 permeability after one month in the Brine 1 increases. Fine needles of xonotlite shows dissolution signs particularly on the core borders.

In area 2, the compositions #2, #3, #4 are investigated. The composition #4 has a density of 1.94 g/cm³ and is cured in the brine 2. Compositions #2 and #3 have a
25 lower density, 1.49 g/cm³. Phases responsible for the cementing properties within this area are mainly anorthite, whatever the slurry density. For the composition #2, a high temperature calcium zeolite, wairakite well detected by XRD and SEM analysis contributes with anorthite to favourable cementitious properties at 300°C. The composition #3, containing more alumina and less silica but having the same density,
30 shows a mineralogical “assemblage” more complex: relic phases as silica and alumina are well detected and wairakite is less observed with SEM. This composition seems to be slower than #2.

If the slurry density is initially higher (#4), anorthite stays the main binder but prehnite, epidote are well detected. Relic phase as silica and alumina are still not completely consumed after one month at 300°C. Wairakite is not detected. Composition #4 contains a part of silica coarser than the one added in the composition #3. That can explain a lower reactivity after one month resulting from lower water saturation in silica useful to form this phase.

After brine attack, whatever the slurry density, anorthite evolves to a sodic plagioclase member, albite that shows well the good adaptation of anorthite to chemical alteration.

The compressive strength and the water permeability of the set cement after curing without or with the brine are presented in table 5.

Table 5: Compressive strength before and after brine attack and water permeability after brine attack for the different compositions.

Formulation Number	#1	#2	#3	#4	#5
Density g/cm ³	1.49	1.49	1.49	1.94	1.94
BHST°F	300	300	300	300	300
Compressive Strength (psi) at BHST after 1 month	900	2332	800	2207	6352
Water permeability (mD)	0.66	0.16	0.2	-	-
Compressive Strength (psi) at BHST after 1 month in brine	830	2500	-	2300	5400
Water permeability (mD) after brine attack	2.81	0.26	-	-	-

Even though the value of the compressive strength is lower for blend #2 than #1 before brine attack, after brine attack, the formulation #2 shows no decrease in the compressive strength. If the brine is directly added as mixing water, this trend is still observed contrary to the trend obtained with the formulation #1 where the compressive strength decrease of 55% of its initial value.

The formulation #2, at a density of 1.49 shows no variations in the compressive strength before and after brine attack and only a slight increase in permeability.

5 The composition #3 exhibits the lower compressive strength with a chemical system not completely reacted.

Electron microscopy gives morphological information that completes the chemical, physical, mechanical and diffractometric data. The set cement matrices provided from formulation #1 and formulation #2 are observed and compared before and after
10 chemical attack. Figures 3a and b, and 4a and b present the set cement matrix obtained with the composition #2. Anorthite is well detected before and after brine attack in the interstitial matrix and crystallizes as interlocking platelets, which can contribute to a superior compressive strength measurement (Figures 3a and b). This phase is also met in some cavities observed in this low-density system. Again
15 anorthite crystallizes under interlocking platelets, plugging the cavity (Figure 4a). Wairakite is well observed in the interstitial matrix under cubic crystals and seems well stable after brine curing (Figure 4b).

Figures 5a and b present the set cement matrix obtained with the composition #1. The
20 samples have been cut in the core near the free surface and inside the core. In view of XRD analysis, xonotlite as fine needles, is the primordial hydrate. Figures 5a and b show the effect of brine curing through the core: in the core near the free surface, the signs of xonotlite dissolution in the interstitial matrix are more significant than inside the core. In view of the SEM observations, the increase of water permeability after
25 brine curing can be explained: 0.66 mD and 2.8 mD before and after brine attack, respectively.

In view of the mechanical and durability results, the composition #2 is a good candidate for geothermal applications. The permeability and the compressive strength
30 are not affected after brine attack at 300°C. Indeed, the results have shown that the faster anorthite forms the lower strength retrogression is observed: at 300°C, the composition #2 exhibits the higher compressive strength with a complete reaction

after one month at high temperature, the composition #3 exhibits the lower compressive strength with a chemical system not completely reacted.

5 The fact that the original material component (cement, cenosphere, aluminum modifiers) have completely reacted can partially explain this durability. Indeed, unreacted silica, and aluminum modifiers react with brine to give secondary phases which are not necessarily stable under these conditions.

10 Anorthite is the calcic member of the solid solution plagioclase that is well known to be stable at high temperature and which can adapt to brine circulation by ions exchanges in its network. Wairakite is a high temperature Ca-zeolite that is able, like all zeolite, to absorb ions in its structure that can be a good property with brine circulation. Its stability domain is well known in the system $\text{Na}_2\text{O}-\text{CaO}-\text{MgO}-\text{Al}_2\text{O}_3-\text{SiO}_2-\text{H}_2\text{O}$ in the range 0-500°C and 01-10kbar involving the phase's anorthite, 15 grossular, prehnite and others. This phase does not exist in formulation #4 at higher density. A fraction of the silica in Composition #4 is coarser (silica) than the silica contained in Composition #3 (silica flour). That can explain a lower reactivity after one month resulting from lower water saturation in silica useful to form this phase. This also shows how the kinetics of the targeted phases (anorthite, wairakite) could be 20 controlled by the particle sizes of the different components of the blend.

Compositions #1 and #2 are also tested to very high temperatures. The set cement cores are cured for one month at 300°C and then cut into smaller samples and cured in a high temperature furnace. They are observed and analyzed by XRD and SEM 25 following different curing durations. Figures 6 – 14 show the results obtained at 300°C (58 days), 600°C (9 days), 750°C (30 days) and 900°C (9 days).

Figure 6 shows a sample of composition #2 after two months at 300°C and can be compared with Figure 7 which shows the prior art composition #1 after the same 30 period at the same temperature. For the prior art composition, cracks and many deleterious signs are observed, whereas composition #2 maintains its integrity well.

Figure 8 shows a sample of composition #2 after one week at 600°C. The set cement sample has been cut again with a small saw. Very consistent slices are obtained and no sign of degradation is observed. This can be compared with composition #1 after the same period at this temperature. The set cement sample is completely
5 disintegrated and friable and it is not possible to saw the sample in the manner of composition #2..

Figures 10 and 11 show compositions #2 and #1 after one week at 750°C, the composition #2 again keeping its integrity contrary to the low-density conventional
10 composition #1.

Figures 12 and 13 show composition #2 after one week at 900°C. Figure 12 shows the whole sample, Figure 13 shows slices of set cement slices cut by a small saw. At 900°C, the new system keeps still a good integrity. The set cement can be cut into
15 slices without splitting. Figure 14 shows a sample of composition #1 after the same period at the same temperature. This sample again shows many cracks and deterioration signs and color variations.

The composition according to the present invention between 300°C and 900°C
20 remains mainly Anorthite. From 600°C, some Wollastonite and Andradite-Grossular garnet minerals are well observed but Anorthite remains the main binder of this cement. Wairakite completely disappears and is replaced by Andradite-Grossular garnets.

25 The very high temperature curing is performed in air inside a refractory furnace and shows the good stability of the new system even at very high temperature. This stability is due to the Anorthite cement which is known to begin to smelt around 1300°C.

30 The present invention provides lightweight formulation using conventional Class G cement for temperatures up to 300°C. The set cement can show very good indications of durability (low permeability and no strength retrogression).

In the process used to design the slurry, the temperature stable phases (anorthite, wairakite) are selected; aluminum modifiers and silica oxides are added to the dry cement so that the elemental composition of the blend corresponds to the phases selected; the control of the particle size distribution and the relative amount of those minerals allow their addition at high concentration while leaving the slurry easily mixable and pumpable; and the kinetics of the targeted phases formation is then controlled by adjusting the crystallinity and the particle sizes of the different solids.

The composition of the oxides of Portland cement is as given in Table 6 (mass %) and of TXI cement is as given in Table 7:

CaO	SiO ₂	Fe ₂ O ₃	Al ₂ O ₃	SO ₃	MgO	Na ₂ O	K ₂ O
64.2	19.4	5.5	4.5	2.8	2.0	0.1	0.6

TABLE 6: Oxides composition of a Portland cement

CaO	SiO ₂	Fe ₂ O ₃	Al ₂ O ₃	SO ₃	MgO	Na ₂ O	K ₂ O
38.5	35.7	5	10.5	Not done	1.8	Not done	Not done

TABLE 7: Oxides composition of a TXI cement

The composition containing such cement has to be enriched with silicon and, optionally, aluminium in order to place it into the target elemental regions. However, the composition has to be able to be prepared with a quantity of mixing water such that the density of the slurry is of the order of the densities usually used for cementing an oil well or the like. Furthermore, the slurry must have rheological properties compatible with pumping in such a well.

The formulations of eleven slurries with densities between 1.89 g/cm³ and 1.49 g/cm³ based on oilwell cement are shown in Table 8 below. For the liquid components (fluid loss control agents and antifoaming agent), the quantities are indicated in gallons per sack of Portland cement (1 gps = 3.78 litres per 94 pound cement sack, i.e., 1 gps = 0.070 litres per kilogram of Portland cement). For all of the other components (the solid components) the quantities are given as the percentages by mass with respect to the mass of Portland cement (BMO = by mass of cement). The position of each composition in the Si-Ca-Al ternary diagram is shown in Figure 16.

All of the proposed formulations are optimised to obtain slurry with a rheology compatible with pumping in a subterranean well, which necessitates adding dispersing agents. Under these conditions, these slurries are stable and pumpable.

- 5 Composition #A is a prior art oil-well cement composition intended for high temperature wells, containing 35% BMOC of silica. The other compositions are compositions of the invention comprising between 0% and 190% BMOC of metakaolin , 0% to 80% BMOC of silica with a grain size close to that of a sand ($D(v,50) = 330 \mu\text{m}$), a Portland cement (average particle size close to $20 \mu\text{m}$), or of
- 10 micro-silica (average particle size close to $1.2 \mu\text{m}$), and 0% to 72% of fly ash having a diameter between $50 \mu\text{m}$ and $400 \mu\text{m}$.

- Further advantageous details and characteristics of the invention will become apparent from the description below of tests established for different examples of additive
- 15 compositions.

Table 8

#	A	B	C	Cbis	D	E	Ebis	F	G	H	I	Ibis
Sand	-	-	-	-	40	-	-	-	-	-	-	-
Silica	35	-	-	-	-	-	37	-	-	-	-	-
Micro-silica	-	80	35.5	52	-	37	-	62	31	-	11.5	11.5
Kaolinite	-	-	-	-	-	-	-	-	-	-	-	28
Metakaolin	-	41	40.4	-	74	88.6	88.6	142	143	142	28	-
Microalumina	-	-	-	14	-	-	-	-	-	-	-	-
Fly ash	-	54.6	44.9	44	14.8	55.2	56.4	72	65	53	31.9	31.9
Dispersing agent	0.3	0.3	0.02	0.03	0.06	0.2	0.15	0.25	0.3	0.3	0.15	0.2
Retarder 1	0.3	0.58	0.46	0.44	0.48	0.59	0.59	0.79	0.71	0.61	0.36	0.4
Retarder 2	0.15	0.36	0.29	0.26	0.3	0.37	0.37	0.49	0.44	0.38	0.22	0.25
Fluid loss control agent	0.6	-	1.2	2.5	-	-	-	-	-	-	-	-
Antifoaming agent	0.05	0.03	0.03	0.03	0.03	0.03	0.03	0.03	0.03	0.03	0.03	0.03

The relative amounts of silicon, aluminium and calcium in these different
5 formulations are as follows:

Table 9

#	A	B	C/Cbis	D	E/Ebis	F	G	H	I/Ibis
Silicon %	46	57	48	49	50	54	50	44	50
Aluminium %	5	18	20	20	25	28	30	32	25
Calcium %	49	25	32	31	25	18	20	24	25

The following table 10 presents the average particle sizes, which have been used with
10 these slurries. These average particle sizes are not a restrictive factor to form Anorthite and can be higher or lower.

Table 10:

#	D(v,0.5)
Silica	330 μm - 2 μm
Kaolinite	10 μm - 2 μm
Metakaolin1	2 μm
Metakaolin2	10 μm
Fly ash	400-50 μm

Table 11 shows the effect of material reactivity, which has to be considered to reach
 5 as soon as possible the anorthite matrix.

The cement slurries are prepared, placed in a reactor and heated for 3 days at 150°C
 and 20 MPa. On the fourth day, the temperature is raised to 300°C over a period of 6
 hours to prevent cracking by thermal shock. The cement slurries are kept under these
 10 conditions for 28 days. Mineralogical analysis by X ray diffraction is shown in Table
 116.

Slurry#Cbis is a prior art slurry containing microalumina and the slurry #C
 metakaolin as aluminium and silicon source. These two blends have a Si/Ca/Al ratio
 15 that falls well within the phase triangle described above.

Table 11:

Formulation	C	Cbis
Metakaolin	-	-
Microalumina	-	-
Quartz	-	-
Xonotlite	*	++
Anorthite	++	++
Wairakite	-	++
Hibonite – 5H	-	-
Riversiderite	+	-

The following symbols are used in Table 11 and in the other tables below reporting
 20 mineralogical analyses:

++: represents a large quantity;

+: represents a small quantity;

*: trace

-: absence

Formulation #C containing metakaolin is clearly more reactive, the set cement matrix after one month is mainly composed by anorthite. With formulation #A, wairakite and xonotlite are still well detected after one month and are metastable.

Table 12 shows the effect of Si/Al/Ca blend composition, the material composition and its size on the anorthite formation at short term. The cement slurries are prepared, placed in a reactor and heated for 3 days at 150°C and 20 MPa. On the fourth day, temperature is raised to 300°C over a period of 6 hours to prevent cracking by thermal shock. The cement slurries are kept under these conditions for 5 days. Mineralogical analysis by X ray diffraction is shown in Table 12:

Table 12:

Formulation	A	B	C	D	E	F	G	H	I	Ibis
Metakaolin	-	-	-	-	-	-	-	-	-	-
Quartz	+	++	-	-	+	++	+	*	+	++
Xonotlite	++	++	+	*	-	-	-	-	-	-
Truscottite	-	++	+	-	+	-	+	-	+	+
Anorthite	-	++	++	++	++	+	++	++	++	+
Diopside	-	-	-	+	-	-	-	-	-	-
Wairakite	-	-	-	-	-	-	-	-	-	-
Hibonite – 5H	-	-	-	++	-	-	-	-	-	-
Riversiderite	-	-	+	-	-	-	-	-	-	-

These results show that whatever the Si/Al/Ca blend composition, included in the Si-Ca-Al triangle in one of the Margarite- Quartz - Pectolite composition triangle, anorthite is well formed in the short term. The nearer the blend composition is to the composition of anorthite, the higher the final mineral assemblage contains anorthite, even in the short term.

The material particle size has an effect on the mineral assemblage at equilibrium at 5 days. But anorthite always is found to be the main binder in the short term, showing

the high reactivity of the coarser material, at high temperature in the Margarite-Quartz-Pectolite composition triangle.

The material nature is a major factor to form quickly anorthite.

5

Table 13 shows the effect of final mineral assemblage after 28 days at 300°C on the set cement properties:

Table 13:

Formulation	3	3bis
Compressive strength after 28 days at 300°C psi (MPa)	2570 (18)	2100 (14.5)
Water permeability (mDarcy)	0.02	0.12

10

Table 11 shows that metakaolin is more reactive than microalumina: the final mineral assemblage is mainly composed of anorthite. Formulation C exhibits after one-month better compressive strength and particularly lower water permeability than formulation Cbis.

15

This water permeability improvement of "Anorthite" cement is an additional positive factor in term of durability towards brine or carbon dioxide.

By selecting as a an aluminium and silicon modifier, an at least partially calcined metakaolin in accordance with the invention, it is possible to obtain a slurry that will set at both bottom hole and surface temperatures, thus avoiding the need to design a specific lead slurry that is optimised to set at surface (i.e. low temperature) and a tail slurry that is optimised to set downhole (i.e. high temperature).

25 Table 14 below shows a comparison between a conventional cement composition (X) optimised to set at high temperature, and a composition (Y) according to the invention.

Table 14:

Slurry	X	Y
Density (g/cm ³)	1.49	1.49
Solids Volume Fraction %	54	57
Free Water	0	0
Blend	Cement Silica Flour Cenospheres	Cement Silica Flour Cenospheres Metakaolin calcination product (PoleStar 200R)
Plastic viscosity (cP)/Yield value after mixing conditioning (lbs/100sqft)	33/6	284/10
Plastic viscosity (cP)/Yield value after 20 min conditioning (lbs/100sqft)	16/13	129/6
Reading on FANN mod.35SA viscosimeter (R1-B1-F1) Gel at 10min	11	14/7
Thickening Time (hours) 300°F (BHCT)	4:00	3:00
UCA Transit Time	0:06	0:04
Compressive Strength psi after 24 hours @ 300°F (BHST)	2200	3450
setting time @ 50 psi	5:30	4:39
setting time @ 500 psi	6:00	4:46
Compressive Strength 24 hours @ 85°F (BHST) 48 hours @ 85°F (BHST) setting time @ 50 psi setting time @ 500 psi	No set after 5 days	114 2125 23:00 27:33

- Sample X is a conventional, low-density slurry (see, for example EP 0 621 247 and
- 5 WO 01/09056) that has been optimised (retarded set) for high temperature conditions by the addition of silica flour. Thickening is observed after 4 hours at BHCT and setting after 5:30/6:00 hours BHST. While a compressive strength of 2200 psi develops after 24 hours at BHCT, no set can be observed at surface temperature. In a
- 10 slurry for the first part of the job (i.e. the portion of the cement that ultimately ends up at the surface) and switch to a retarded slurry for the second part of the job (i.e. the portion of the cement that ultimately is exposed to the high temperatures). Obviously

there are problems in such an approach. Switching between two slurries with such differing behaviours can present difficulties if there is any mixing between the two slurries in the well. Also, it is inevitable that some part of each slurry will be exposed to conditions for which it is not optimised with the possibility that it does not set.

5

Sample Y is a slurry according to one embodiment of the invention. The modification of silicon and aluminium composition is achieved using silica flour and a calcination product of metakaolin (PoleStar 200R). The effect of this is that not only a good high temperature set observed (thickening at 3 hours, setting at 4:39/4:46 hours BHST, compressive strength of 3450 psi after 24 hours BHST) but setting is also observed at surface temperature (23:00/27:33 hours, compressive strength after 24 hours 114psi and after 48 hours 2125 psi). Thus single cement slurry can be used to cement the whole well, avoiding the problems with using different lead and tail slurries discussed above. Particular benefits are: saving of rig time, improved mud removal with reduction of risk of vapour pressure induced casing collapse.

10
15

While the calcination product of metakaolin used in this example is PoleStar 200R, other such materials can be used. A number of metakaolin products with greater or lesser degrees of calcinations are readily available and can be used depending on the effect desired.

20

The methods of the invention can be used to design cement blends of a suitable cement and other solid additives which, when mixed with sufficient water can form pumpable slurries. In well cementing, the slurries are pumped into the well in question and allowed to set to provide zonal isolation and well stabilization. The compositions can also be used to line pipes. In this case, a cement slurry is placed inside a pipe, which is then rotated, typically at high speed to coat the slurry onto the inside of the pipe where it is allowed to set.

25

CLAIMS

- 1 A method of designing a cement slurry, comprising:
- 5 i) determining a temperature to which the cement slurry will be exposed in situ;
- ii) determining a stable, thermodynamic equilibrium composition of a $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2\text{-H}_2\text{O}$ (CASH) mineral system, analogous to the cement when set, at the determined temperature;
- 10 iii) determining proportions of cement and mineral oxides required to provide a mixture having the determined composition of the CASH mineral system; and
- iv) defining a series of particulate materials of predetermined particle sizes and densities, and comprising the cement and mineral oxides in the determined proportions such that, when mixed with water, a pumpable
- 15 slurry is formed.
- 2 A method as claimed in claim 1, wherein the mineral system is a $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2\text{-H}_2\text{O}$ (CASH) mineral system.
- 20 3 A method as claimed in claim 1 or 2, wherein the cement is a Portland or alumina type cement.
- 4 A method as claimed in any of claims 1 – 3, wherein the series of particulate materials is defined so as to promote the formation of anorthite.
- 25 5 A method as claimed in any preceding claim, wherein the series of particulate materials is selected to provide a pumpable slurry of density less than 1.94.
- 6 A method as claimed in claim 5, wherein the step of defining the series of
- 30 particulate materials includes selecting a low density particulate component having a density of less than 0.8 sg as part of the series.

- 7 A method as claimed in any preceding claim, wherein the step of defining a series of particulate materials includes the step of defining a series of discrete particle size fractions to optimise the pacing of the solid particles in the slurry.
- 5 8 A cement composition comprising a mixture of particulate materials comprising cement and a mixture of mineral oxides, the mineral composition of the mixture comprising a stable thermodynamic equilibrium composition of a CASH mineral system at the desired temperature of use, the particulate materials being present in predetermined particle size ranges and densities.
- 10
- 9 A cement composition as claimed in claim 8, wherein the mixture comprises one or more mineral oxides that contribute Si, Ca, or Al to the mixture so as to place its mineralogical composition within the Margarite-Quartz-Pectolite region of the Si-Ca-Al phase diagram, and optimise early formation of
- 15 Anorthite.
- 10 A cement composition as claimed in claim 9, wherein the one or more mineral oxide contributing Si, Ca, or Al comprise kaolin, calcination products of kaolin, metakaolin, and/or calcination products of metakaolin.
- 20
- 11 A cement composition as claimed in claim 10, wherein the mineral oxides comprise:
- up to 190% BMOC kaolin, calcination products of kaolin, metakaolin, and/or calcination products of metakaolin;
 - 25 - up to 80% BMOC silica; and
 - up to 72% BMOC fly ash.
- 12 A cement composition as claimed in claim 11, wherein the average particle size of the kaolin, calcination products of kaolin, metakaolin, and/or
- 30 calcination products of metakaolin is 2µm to 10µm, of silica is 2µm to 330µm, and of fly ash is 50µm to 400µm.

- 13 A cement composition as claimed in any of claims 8 – 12, wherein the mixture has a Si-Ca-Al composition comprising Si 44% - 57%, Ca 18% - 32%, and Al 18% - 32%.
- 5 14 A cement composition as claimed in any of claims 8 – 13, further comprising a low density particulate material comprising silica/alumina microspheres.
- 15 A cement composition as claimed in any of claims 8 - 14, further comprising sufficient water to form a pumpable slurry.
- 10 16 A cement composition as claimed in claim 15, wherein the slurry has a density of 1.3 – 1.94 S.G..
- 17 A method of cementing a well comprising pumping a slurry as claimed in claim 15 or 16 into a well.
- 15 18 A method as claimed in claim 17, further comprising:
- 20 i) designing a cement slurry according to a method as claimed in any of claimed 1 – 7 such that the temperature of the well is used as the determined temperature;
- 20 ii) forming a slurry from the defined series of particulate materials and water; and
- 20 iii) pumping the slurry into the well.
- 25 19 A method as claimed in claim 15, 16 or 17, wherein the well has a relatively high temperature region and a relatively low temperature region, the cement composition comprising materials necessary to form the stable thermodynamic equilibrium composition of a CASH mineral system at the relatively higher temperature, the method comprising pumping a single slurry into both the relatively high temperature region and a relatively low temperature region.
- 30 20 A method as claimed in any of claims 15 – 19, wherein the well has a maximum temperature in the range 250°C – 900°C.

- 21 A method of lining a pipe for high temperature fluids, comprising:
- 5 i) preparing a cement slurry as claimed in claim 15 or 16, the determined temperature being the temperature of the fluids in the pipe;
- ii) placing the slurry in the pipe; and
- iii) rotating the pipe so as to deposit the slurry evenly on the inner surface of the walls of the pipe.

10

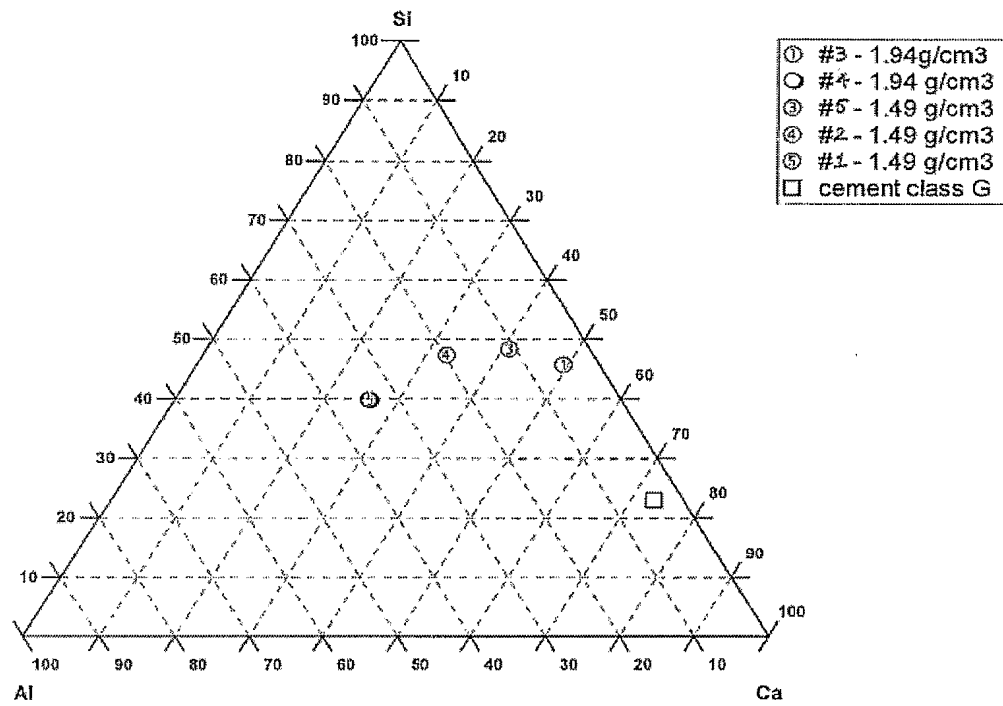


FIGURE 1

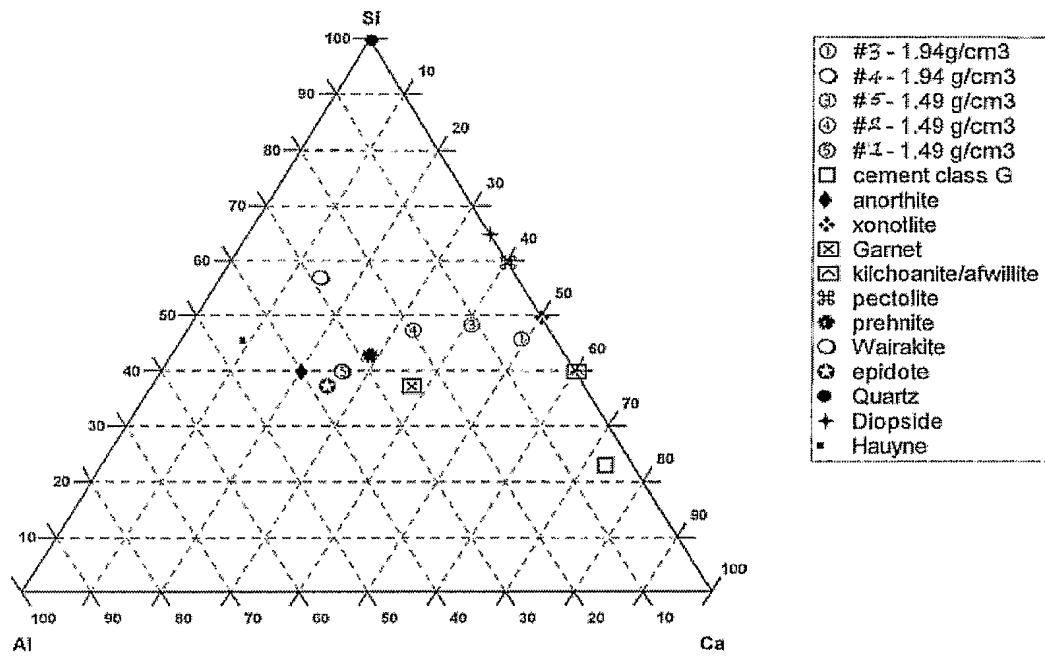


FIGURE 2

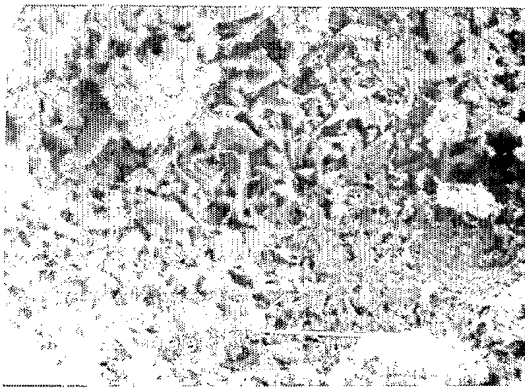


FIGURE 3a

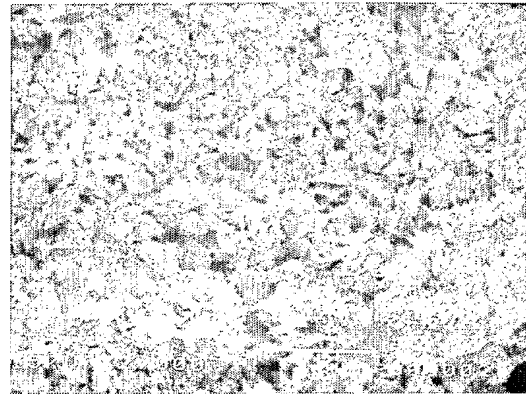


FIGURE 3b

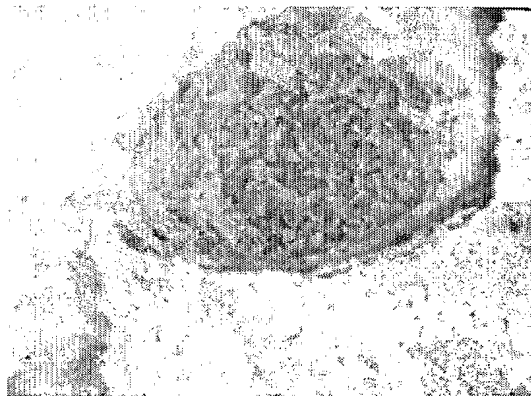


FIGURE 4a

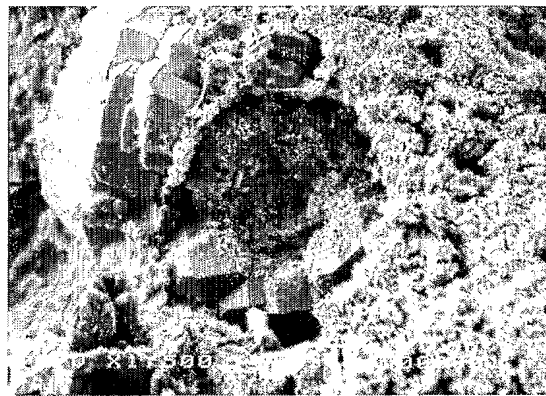


FIGURE 4b

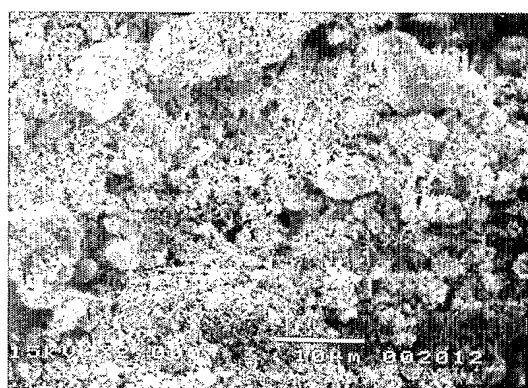


FIGURE 5a

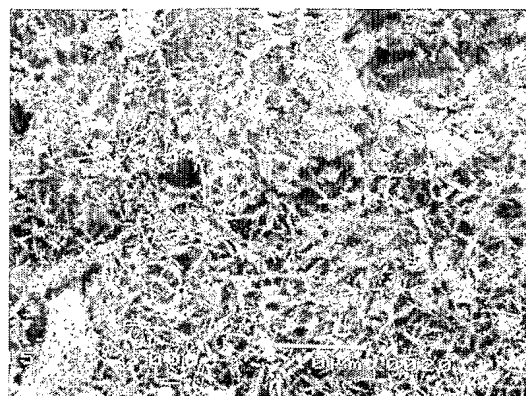


FIGURE 5b

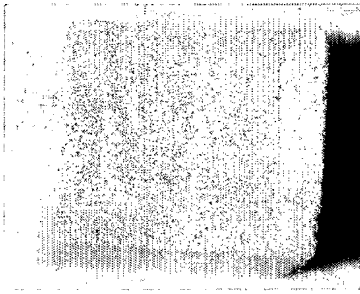


FIGURE 6



FIGURE 7

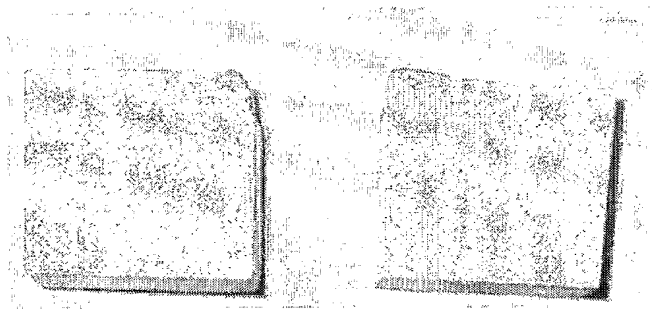


FIGURE 8



FIGURE 9

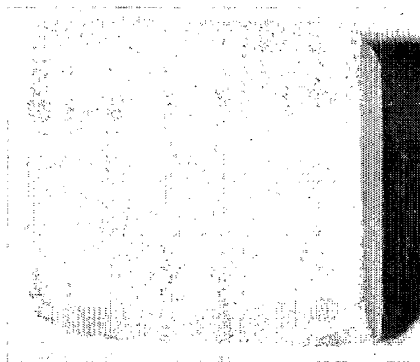


FIGURE 10



FIGURE 11

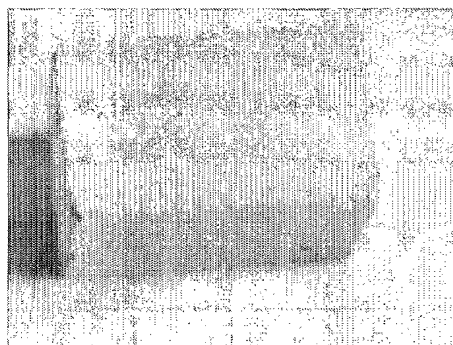


FIGURE 12



FIGURE 13

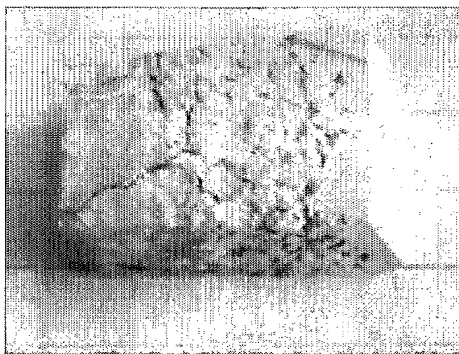
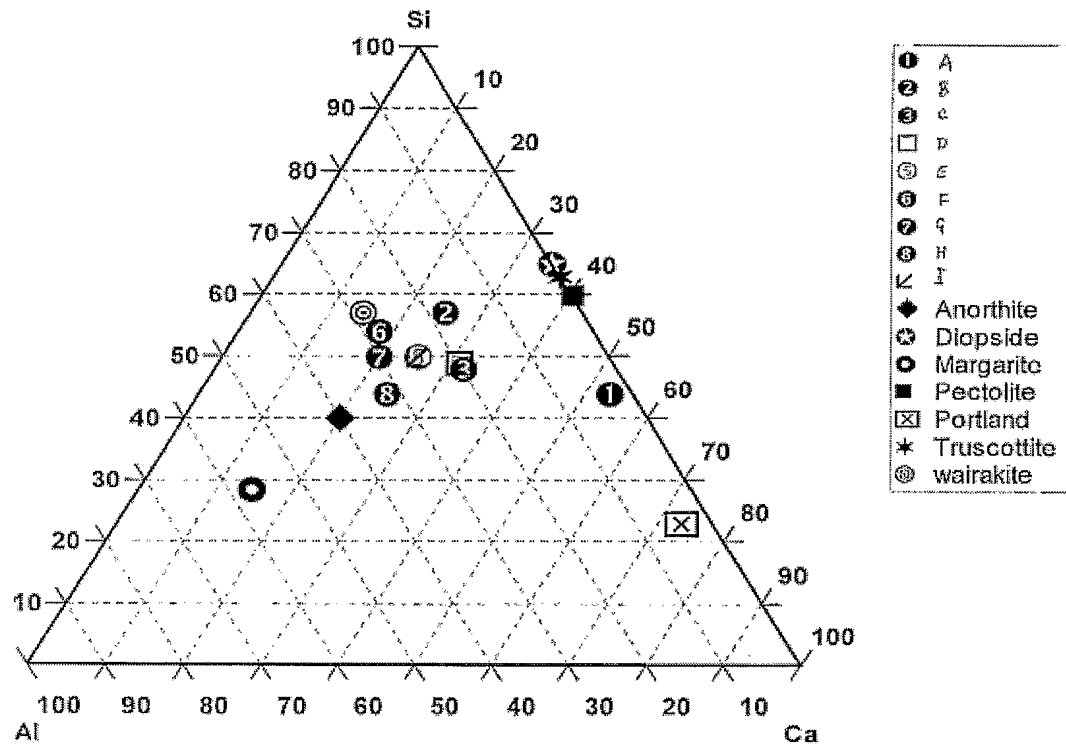


FIGURE 14

FIGURE 15



INTERNATIONAL SEARCH REPORT

International Application No
PCT/EP 03/01578

A. CLASSIFICATION OF SUBJECT MATTER
IPC 7 C04B40/00

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)
IPC 7 C04B

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practical, search terms used)

WPI Data, PAJ, EPO-Internal, COMPENDEX

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	WO 01 70646 A (SOFITECH NV ;SCHLUMBERGER CA LTD (CA); BARLET GOUEDARD VERONIQUE () 27 September 2001 (2001-09-27) cited in the application claims ----	1-3,5-9, 13-21
Y	EP 0 621 247 A (SOFITECH NV) 26 October 1994 (1994-10-26) cited in the application page 7, line 19; claim 1 ----	1-3,5-9, 13-20
Y	WO 98 33749 A (CONROY PAUL JAMES ;NEWMAN JOHN BRIAN (GB); WRC PLC (GB); HUDSON LI) 6 August 1998 (1998-08-06) claims 20,24,25,29 ----- -/--	21

☒ Further documents are listed in the continuation of box C.

☒ Patent family members are listed in annex.

* Special categories of cited documents :

- *A* document defining the general state of the art which is not considered to be of particular relevance
- *E* earlier document but published on or after the international filing date
- *L* document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)
- *O* document referring to an oral disclosure, use, exhibition or other means
- *P* document published prior to the international filing date but later than the priority date claimed

- *T* later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
- *X* document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone
- *Y* document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.
- * & * document member of the same patent family

Date of the actual completion of the international search

20 May 2003

Date of mailing of the international search report

27/05/2003

Name and mailing address of the ISA
European Patent Office, P.B. 5818 Patentlaan 2
NL - 2280 HV Rijswijk
Tel. (+31-70) 340-2040, Tx. 31 651 epo nl,
Fax: (+31-70) 340-3016

Authorized officer

Daeleman, P

INTERNATIONAL SEARCH REPORT

International Application No
PCT/EP 03/01578

C.(Continuation) DOCUMENTS CONSIDERED TO BE RELEVANT

Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	WO 99 01397 A (SOFITECH NV ;SCHLUMBERGER CA LTD (CA); BARLET GOUEDARD VERONIQUE () 14 January 1999 (1999-01-14) cited in the application claims ----	1-20
A	US 5 553 670 A (COWAN KENNETH M) 10 September 1996 (1996-09-10) claims 6,9 ----	1,2
X	WO 99 12860 A (NORTH AMERICAN REFRACTORIES) 18 March 1999 (1999-03-18) page 2, line 30-33 page 20, line 22-29; claim 24 ----	8
A	----	9-13,15
A	WO 02 10086 A (GHOFRANI REZA ;RADU GABRIELA (DE); ROGOJINOIU EMIL (DE); MIEHE HEI) 7 February 2002 (2002-02-07) ----	
A	WO 95 05350 A (KHASHOGGI) 23 February 1995 (1995-02-23) claim 1 ----	
A	US 6 145 591 A (V. BONCAN, ET AL.) 14 November 2000 (2000-11-14) claim 9 ----	
A	WO 95 19942 A (BJ SERVICES) 27 July 1995 (1995-07-27) ----	
A	FR 2 384 729 A (VIZGAZDALKODASI TUDOMANYOS KUTATO KOZPONT) 20 October 1978 (1978-10-20) -----	

INTERNATIONAL SEARCH REPORT

International Application No
PCT/EP 03/01578

Patent document cited in search report		Publication date	Patent family member(s)	Publication date
WO 0170646	A	27-09-2001	FR 2806717 A1 AU 5040201 A WO 0170646 A1 EP 1268361 A1	28-09-2001 03-10-2001 27-09-2001 02-01-2003
EP 0621247	A	26-10-1994	FR 2704218 A1 AU 666261 B2 AU 6053094 A BR 9401553 A CA 2117276 A1 DE 69419352 D1 DE 69419352 T2 DK 621247 T3 EP 0621247 A2 GB 2277927 A , B JP 7144952 A NO 941440 A	28-10-1994 01-02-1996 27-10-1994 22-11-1994 22-10-1994 12-08-1999 13-01-2000 13-12-1999 26-10-1994 16-11-1994 06-06-1995 24-10-1994
WO 9833749	A	06-08-1998	EP 1012125 A1 WO 9833749 A1	28-06-2000 06-08-1998
WO 9901397	A	14-01-1999	FR 2765571 A1 AU 8540498 A DE 69805594 D1 DE 69805594 T2 DK 922013 T3 WO 9901397 A1 EP 0922013 A1	08-01-1999 25-01-1999 04-07-2002 16-01-2003 23-09-2002 14-01-1999 16-06-1999
US 5553670	A	10-09-1996	NONE	
WO 9912860	A	18-03-1999	US 5976240 A BR 9814944 A CN 1269772 T EP 1027301 A1 JP 2001515834 T TW 473461 B WO 9912860 A1	02-11-1999 31-07-2001 11-10-2000 16-08-2000 25-09-2001 21-01-2002 18-03-1999
WO 0210086	A	07-02-2002	DE 10037118 A1 AU 7956301 A WO 0210086 A1 EP 1305263 A1 NO 20030414 A	14-02-2002 13-02-2002 07-02-2002 02-05-2003 27-01-2003
WO 9505350	A	23-02-1995	AU 679784 B2 AU 7670994 A BR 9407168 A CA 2168643 A1 CN 1100395 A EG 20631 A EP 0714383 A1 IL 110605 A JP 8511486 T NZ 273435 A RU 2135427 C1 WO 9505350 A1 US 6030673 A	10-07-1997 14-03-1995 17-09-1996 23-02-1995 22-03-1995 31-10-1999 05-06-1996 16-08-1998 03-12-1996 24-10-1997 27-08-1999 23-02-1995 29-02-2000

INTERNATIONAL SEARCH REPORT

Inter al Application No
PCT/EP 03/01578

Patent document cited in search report	Publication date	Patent family member(s)	Publication date
WO 9505350	A	US 5549859 A	27-08-1996
		US 5695811 A	09-12-1997
		US 5660900 A	26-08-1997
		US 5783126 A	21-07-1998
		US 5545297 A	13-08-1996
		US 5662731 A	02-09-1997
		US 5683772 A	04-11-1997
		US 5709827 A	20-01-1998
		US 5679145 A	21-10-1997
		US 5527387 A	18-06-1996
		US 5658624 A	19-08-1997
		US 5798151 A	25-08-1998
		US 5868824 A	09-02-1999
		ZA 9405497 A	07-06-1995
		ZW 10394 A1	28-09-1994
US 6145591	A	14-11-2000	NONE
WO 9519942	A	27-07-1995	US 5447197 A
			AT 190298 T
			DE 69423370 D1
			DE 69423370 T2
			DK 741679 T3
			EP 0741679 A1
			NO 963103 A
			WO 9519942 A1
			US 5547506 A
FR 2384729	A	20-10-1978	FR 2384729 A1